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Improved reset breakdown strength in a HfO$_x$-based resistive memory by introducing RuO$_x$ oxygen diffusion barrier

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We investigated the reset breakdown phenomenon of HfO$_x$-based resistive memory for reliable switching operation in a fully CMOS compatible stack. Through the understanding on the effect of electrode materials and device area, our findings show that observed failure is attributed to additional oxygen vacancies close to the electrode interface, where switching is occurred. Therefore, RuO$_x$ serving as an oxygen diffusion barrier was introduced to suppress the generation of unwanted oxygen vacancies by preventing out-diffusion of oxygen through the electrode. As a result, significantly enhanced breakdown strength in HfO$_x$/RuO$_x$ stack is achieved and resulting in improved cycle endurance with larger on/off ratio.

INTRODUCTION

Resistive random-access memory (RRAM) has been considered as one of the promising candidates for NAND FLASH memory because of the low power operation (~ pJ), high scalability (sub-10 nm), and CMOS compatibility. These attractive characteristics of RRAM are attributed to the physical mechanism, which has been known as the redox reaction of oxygen vacancy in various transition-metal oxides including TiO$_2$, Ta$_2$O$_5$ and HfO$_2$. Owing to the transition between formation and rupture of a filament at oxide/electrode interface, low-resistance state (LRS) and high-resistance state (HRS) can be reversibly obtained.

Although, the use of localized filament accompanying the migration of oxygen vacancies in RRAM is beneficial to the switching characteristics, stochastic movement of oxygen vacancies causes performance degradation in reliability test such as data retention, switching uniformity, and cycling endurance. Not only that, it has been recently reported that unexpected permanent breakdown was induced when the device was switched from LRS to HRS i.e. during the reset process. After the breakdown, the device stuck in LRS and further resistance switching could not be obtained. Based on the literature, this failure could be contributed by the amount of oxygen vacancies at the bottom electrode (BE), which was not involved in switching operation. However, the effect of BE conditions on the breakdown behavior has not yet been fully understood.

Therefore, in this study, the reset breakdown phenomenon in HfO$_x$-based RRAM was systematically investigated by considering the reactivity of BE materials and the size of active device area. Our result revealed that the device failure with low breakdown voltage was mainly observed when the considerable amount of oxygen vacancies at HfO$_x$/BE interface were induced. Hence, RuO$_x$ layer serving as an oxygen diffusion barrier was inserted into BE interface to minimize the

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formation of unwanted oxygen vacancies, and the reliable switching characteristics with enhanced breakdown strength were achieved.

EXPERIMENTAL PROCEDURES

The HfO$_x$-based RRAM device was fabricated on W-substrate with various device diameters of 200 nm, 500 nm, 700 nm, and 1 um. A 4 nm-thick HfO$_x$ switching layer was deposited by atomic layer deposition (ALD) technique at 250°C. For comparison of effect of electrodes, 100-nm-thick films of various materials of Pt, Ru, and Ta were deposited by using sputtering system at room temperature. For the deposition of 60-nm-thick RuO$_x$ layer, reactive sputtering was used with Ru metal target in an Ar (30 sccm) and O$_2$ (3 sccm) ambient.

RESULTS & DISCUSSION

Note that we intentionally applied positive bias on the W-substrate to clarify the effect of BE on the breakdown behavior. Thus, the TE was fixed as W-substrate, and various electrode materials can be evaluated as BE hereafter, as shown in inset of Fig. 1(a). The switching current in all the samples were identically limited at 300 uA to prevent the damage into the cells due to overflowing current. Fig. 1(a) shows the typical I-V characteristics of W/HfO$_x$/BE stack with 1 um$^2$ active area. It is obvious that the typical resistive switching characteristics were observed irrespective of the BE materials, while the breakdown behavior was only occurred in the Ta BE device during reset process close to the reset voltage. To understand the relation between BE and breakdown behavior, we extracted breakdown voltage ($V_{BD}$) from the DC I-V data with increased stop voltage, as shown in Fig. 1(b). The current gradually decreased by applying RESET voltage, and then sudden increase in the current was observed at higher voltage regime. The average value of $V_{BD}$ as a function of BE materials clearly indicate that the higher the oxygen affinity of BE, the smaller is the $V_{BD}$. This $V_{BD}$ trend can be understood by considering Gibbs free energy ($\Delta G$) of formation of oxide, which is

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**FIG. 1.** (a) The I-V characteristics of W/HfO$_x$/Pt (Pt BE), W/HfO$_x$/Ru (Ru BE), and W/HfO$_x$/Ta (Ta BE) devices after forming process. In Ta BE device, the breakdown failure is shown in 1.2V. (b) The breakdown voltage ($V_{BD}$) of Pt BE and Ru BE device. (c) Distribution of $V_{BD}$ values in each device.
FIG. 2. Measurement of $V_{BD}$ with device area in (a) Ta BE device, (b) Pt BE device. The $V_{BD}$ is increased with decrease of device area in Pt BE device.

generally represented as the index of redox reaction.\textsuperscript{14} The lower absolute value of $\Delta G$ value favors lower redox reaction at electrode/oxide interface. Therefore, the Ta electrode which has $\Delta G$ value of -760 kJ/mol can absorb more oxygen than that of Ru and Pt electrodes which have $\Delta G$ values of -213 kJ/mol and -164 kJ/mol respectively.\textsuperscript{15–17} This means that a lot of oxygen vacancies can be easily generated in Ta BE compared to Pt and Ru BE.

The lower $V_{BD}$ obtained in the device having a large amount of oxygen vacancies close to BE was further verified by examining the effect of the device area on the $V_{BD}$ as shown in Fig. 2(a) and 2(b). For the device with reactive Ta BE, breakdown around 1.5 V was observed regardless of active area. On the other hand, when active area was reduced from 1 um to 200 nm for inert Pt BE, the $V_{BD}$ was increased with respect to area scaling.

Based on these results, a plausible model can be proposed to explain the varying $V_{BD}$ with respect to the reactivity of BE materials and device area, as shown in Fig. 3. It is generally explained that oxygen vacancies involved in the switching operation are mainly generated from reactive TE/oxide interface. The formed oxygen vacancies can be thus migrated towards bottom electrode.

FIG. 3. The schematic diagram of breakdown failure process Ta BE device (left) and Pt BE device (right).
(BE) when we apply positive bias at TE, and then these vacancies form a conductive filament inside the oxide switching layer, which corresponds to LRS. On the other hand, the HRS can be achieved by drifting the oxygen vacancies apart from the filament close to BE under the smaller negative bias. It is worthy to note that this working scenario has been explained in case of a RRAM device having inert BE material such as Pt. When the RRAM device was fabricated on top of non-inert BE such as TiN, it should be considered for the generation of additional oxygen vacancies from the BE interface also.\textsuperscript{12,13} Thus, when the oxygen vacancies start to move away from the filament under the larger negative bias, additionally formed oxygen vacancies nearby BE simultaneously can move toward the ruptured regime. This re-connection of the filament in the HRS was the reason why the breakdown behavior was more easily observed in the reactive BE materials. In addition, when the active area was smaller, the probability of the generation of oxygen vacancies was decreased. Thus, the breakdown strength in the inert BE with a smaller active area was further enhanced, as shown in Fig. 2(b). Based on the above results, inert metal electrode such as Pt should be considered to prevent the breakdown failure during reset operation due to the low oxygen vacancy quantity. However, the inert materials cannot be utilized in CMOS process because of etching and cost issues. While the tuning of bias conditions by applying small reset bias can also be used as an alternative approach, but it causes degraded switching on/off ratio.

Here, we suggested that the RuO\textsubscript{x} layer was introduced to suppress the effect of scavenging on top of the HfO\textsubscript{x} layer for fully CMOS compatible stack. The RuO\textsubscript{x} material is widely used as an oxygen diffusion barrier to prevent the out-di ffusion of oxygen through the electrode in ferroelectric memory application.\textsuperscript{18,19} The I-V characteristic of the HfO\textsubscript{x}/RuO\textsubscript{x} system with 200nm device is shown in Fig. 4(a). The stable switching and high on/off ratio was obtained after 100 dc consecutive cycles. To confirm the breakdown phenomenon, the dc voltage sweep was conducted from 0V to 5V. The result is that the breakdown did not occur even for >5 V bias and it was similar to that of

![Image](image.png)

**Fig. 4.** (a) The I-V characteristic of the W/HfO\textsubscript{x}/RuO\textsubscript{x} (RuO\textsubscript{x} BE) device with 1, 10, and 100 cycles and evaluation of the breakdown failure of RuO\textsubscript{x} BE device. The breakdown failure was not shown in RuO\textsubscript{x} BE device. (b) AC pulse endurance and (c) the LRS and HRS resistance distribution of RuO\textsubscript{x} BE device with 1us pulse width. The uniform resistance switching was obtained without the degradation of memory characteristic up to 10\textsuperscript{6}. 
FIG. 5. Chemical bonding states of the O 1s states of (a) Ta BE device and (b) RuOₓ BE at HfOₓ/electrode interface.

the device with Pt BE. This observed robust switching behavior is beneficial not only for switching reliability but also for the possibility of multi-level cell due to the gradually increased resistance, which was controlled by the reset bias. Fig. 4(b) and 4(c) show the pulse endurance property of the RuOₓ BE device. The pulse switching shows high memory margin of >10² and switching of >10⁶ cycles with 1us pulse width. In addition, the uniform 1000 consecutive switching cycle distribution was obtained as shown in Fig. 4(c).

We believed that this robust switching behavior was related to the restricted generation of oxygen vacancies by preventing the oxygen diffusion toward RuOₓ from HfOₓ. To analyze the oxygen state of the HfOₓ/electrode interface, the X-ray photoelectron spectroscopy (XPS) was performed. Fig. 5(a) and 5(b) show the O 1s spectra of the HfOₓ layer in Ta BE and RuOₓ BE devices. According to the XPS result, the concentration of lattice oxygen is larger than the concentration of non-lattice oxygen in the RuOₓ BE device. In contrast, the concentration of lattice oxygen is similar with the concentration of non-lattice oxygen in the HfOₓ/Ta interface. The results indicated that the low oxygen vacancy quantity in the HfOₓ/RuOₓ interface.

CONCLUSION

In this study, the reset breakdown phenomenon in a HfOₓ-based RRAM was investigated by means of various BE materials and device sizes. Our findings show that the large amount of oxygen vacancies at the BE interface, which could be generated by either scavenging effect of reactive electrode material or large active area, induced a permanent breakdown behavior. Thus, we introduced the RuOₓ layer, serving as oxygen diffusion barrier, to minimize the formation of oxygen vacancies at the BE interface. This CMOS compatible HfOₓ/RuOₓ system allows the dramatically enhanced breakdown strength with robust switching characteristics.

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