High-Quality Cobalt Thin Films by Plasma-Enhanced Atomic Layer Deposition

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High-quality Co films with low resistivity (10 μΩ cm) were deposited by plasma-enhanced atomic layer deposition (PE-ALD) from metalorganic precursors and NH3 plasma. The deposition characteristics and film properties were investigated. Especially, we compared the results using two cyclopentadienyl Co precursors, CoCp(2) and CoCp2. While low resistivity Co films were deposited by both precursors, much better self-limiting behavior was observed for CoCp2. Rutherford backscattering and X-ray photoelectron spectroscopy analysis have shown that the impurity contents in PE-ALD Co film were very low. CoSi2 formation by post deposition annealing with Ti capping layer was studied by synchrotron X-ray diffraction.

Figure 1. XPS depth profile of thermal ALD Co on Si(001) substrate from CoCp(2) and H2 showing the Co, C, O, and Si contents in the film.

The contact resistance increase with decreasing line width for the source/drain contact using TiSi2 as a contact material poses a serious problem with device scaling.2,3 CoSi2 has been studied as an alternative due to its immunity to the shrinkage of line width, low resistivity, and thermal stability.2,3 In other words, the sheet resistance of CoSi2 remains almost constant with scaling down while that of TiSi2 steeply increases. The effect is routinely mentioned as a fine line effect.2 Sputtering has been a standard deposition technique for Co, which is annealed to form CoSi2 contact through self-aligned silicide process. In current dynamic random access memory (DRAM) technology, stacked capacitor structure has benefits over trench capacitor for abiding by the scaling down of the memory devices.5 For stacked capacitor structure, however, the capacitor-over-bitline (COB) requires that the contact material should be formed in deep contact holes with high aspect ratio. Thus, inherent poor step coverage of sputtering is becoming more problematic for sub-50 nm technology node high density DRAM fabrication.6

Atomic layer deposition (ALD) is a promising deposition technique in the nanoscale regime due to its excellent conformality and thickness controllability at atomic scale. However, ALD of Co has rarely been reported except for recent reports using laboratory-synthesized acetamidinate precursor, without detailed film properties.8,9 The ALD of high-quality Co using commercially available Co precursors is important for the implementation of Co ALD for nanoscale device contact fabrication. In this study, we developed a plasma-enhanced ALD (PE-ALD) process using metalorganic (MO) precursors and NH3 plasma. High purity Co thin films with low resistivity, close to the bulk value, were successfully deposited using various metalorganic Co precursors, including bis-cyclopentadienyl Co (CoCp(2)) and cyclopentadienyl dicarbonyl Co (CoCp(2)CO2). We compared the growth characteristics and film properties of PE-ALD from these two cyclopentadienyl Co precursors.

A remote plasma-enhanced ALD system was built and used in this study. A detailed configuration of the chamber can be found in our previous report.10 To produce adequate vapor pressure, the temperature of the bubbler containing CoCp2 (solid) was maintained at 78°C, while a cooling unit was used for CoCp(2)CO2 (liquid) to keep the temperature at 0°C to avoid excessive vapor pressure buildup. The reactant gas, NH3, was flown into the chamber through remote plasma generation system located at the center of the wafer for obtaining better uniformity. The plasma generator is composed of a quartz tube wrapped by Au coated stainless steel coil and auto matching network. The standard PE-ALD process was composed of precursor exposure time \( t_p = 2 \) s and reactant exposure time \( t_r = 6 \) s with purging time \( t_g = 3 \) s between these two steps. The typical growth temperature, \( T_g \), was 300°C. NH3 flow and plasma power were constant for all the experiments at 200 sccm and 300 W for CoCp2. Similar growth conditions were used for CoCp(2)CO2. A sputtering chamber, connected to PE-ALD chamber through load-lock, was used for deposition of Ti capping layer on PE-ALD Co for postdeposition thermal annealing. Postdeposition annealing was done at rapid thermal annealing (RTA) system in N2 environment.

Si(001) and SiO2 were used as substrates. The Si substrate was cleaned by dipping in diluted HF solution (1%, for 30 s) followed by dry N2 blowing and loaded into the chamber immediately to prevent the formation of native oxide. The SiO2 substrates were cleaned by dipping in acetone, isopropyl alcohol, and deionized DI water sequentially, followed by N2 blowing. The thickness of the films was routinely measured by profilometer or scanning electron microscopy (SEM) and the resistivity by four-point probe. Synchrotron radiation X-ray diffraction (Pohang Light Source, 3C2 beam line) with 2° glancing angle was used for the microstructure analysis of Co and CoSi2 films. The chemical bonding and impurity level of Co films were investigated by Rutherford backscattering (RBS), X-ray photoelectron spectroscopy (XPS, Al Kα), and secondary ion mass spectrometry (SIMS). The samples were in situ cleaned by Ar sputtering right before XPS analysis.

At initial experiments, we tried thermal ALD of Co using CoCp(2) as a precursor and H2 as a reactant. Irrespective of the substrates, however, highly carbon contaminated amorphous Co-C films were deposited. Figure 1 shows the depth profile by XPS of thermal ALD Co deposited on Si(001) at H2 flow of 200 sccm. It is shown that the carbon content in the film was as high as 50 atom % and the resistivity was over 2000 μΩ cm. Moreover, the XRD results have shown that the thermal ALD Co has amorphous structure. Based on the XPS and XRD results, we can infer that the thermal
ALD Co film is amorphous Co-C. The carbon might be incorporated as hydrocarbon, which requires more analysis of the films. The growth rate increases almost monotonically with increasing precursor exposure time indicating that the saturation adsorption of CoCp(CO)₂ molecules does not occur. Also, Co-C thin film was deposited even without any reactant at T = 300°C. Similarly, for thermal ALD of Co using CoCp₂ and H₂, highly C-contaminated amorphous phase was deposited on SiO₂ substrate. In contrast, however, no deposition occurred on HF-dipped Si substrate. Co chemical vapor deposition (CVD) using CoCp(CO)₃ or CoCp₂ was reported to produce highly carbon contaminated Co on SiO₂ and Si₃N₄ surfaces when the hydrogen pressure during deposition was low, because the carbon was incorporated into the film through disproportionation reaction. Thermal disproportion of C₅H₅ results in unstable and nonvolatile cyclopentadienyl radical such as dihydrofulvalene leading to high C incorporation. Meanwhile, on H-terminated Si surface, very low reactivity of CoCp₂ causes nucleation problem resulting in no deposition, which agrees with our result.

To effectively remove Cp ligands without carbon incorporation, we deposited Co by PE-ALD using NH₃ plasma as a reactant. This process produced metallic, low resistivity films at a temperature range of 250-400°C on both Si and SiO₂, irrespective of Co precursors. In contrast to thermal ALD producing amorphous phase, XRD analysis on PE-ALD Co thin films has clearly shown diffraction peak at 44.2°, which is indexed as β-Co(111) peak. High-resolution transmission electron microscopy analysis also has shown that the PE-ALD Co film is composed of polycrystalline grains. RBS and SIMS were used to analyze the film composition of PE-ALD Co. Besides Co, no impurity related feature was observed for PE-ALD Co films prepared at T = 300°C on both standard RBS. Oxygen or nitrogen was not detected even with oxygen and nitrogen enhanced measurement. Only small carbon peak was observed by carbon resonance enhanced RBS, mostly due to the surface contamination. The purity of the PE-ALD Co films was also confirmed by SIMS measurement. The N and O concentrations in the PE-ALD Co films were even lower than those in sputtered Co film, which was prepared as a reference.

More detailed compositional analysis including chemical bonding state was carried out by XPS. Figure 2 shows the typical XPS spectra of PE-ALD Co from CoCp(CO)₂ prepared at T = 300°C. Figure 2a shows that the binding energy difference between Co 2p₃/2 peak and 2p₁/2 peak is 14.97 eV and the peak positions of Figure 2a shows that the binding energy difference between Co 2p₃/2 peak and 2p₁/2 peak is 14.97 eV and the peak positions of

![Figure 2](image)

Figure 2. XPS spectra of PE-ALD Co on Si(001) substrate from CoCp(CO)₃ and NH₃ plasma showing the energy range for (a) Co 2p, (b) O 1s, (c) N 1s, and (d) C 1s.

Figure 3 shows the growth characteristics of PE-ALD Co from CoCp(CO)₂ (Fig. 3a) and CoCp₂ (Fig. 3b) with NH₃ plasma at growth temperature T = 300°C on SiO₂. Almost the same results were obtained on Si. The growth rate vs CoCp(CO)₂ exposure time, t, shows apparent saturation at t = 1 ~ 3 s range, but rapidly increases at longer exposure time. This indicates that the CoCp(CO)₂ adsorption does not occur with good saturation. Also, as mentioned above, the ALD using CoCp(CO)₂ was found to produce highly carbon contaminated film with high resistivity even without reactant. These observations can be explained by low thermal stability of CoCp(CO)₂. The CoCp(CO)₂ is reported to be thermally decomposed at as low temperature as 140°C. This low thermal stability is probably because the carbonyl ligand is easily removed from CoCp(CO)₂ by thermal energy. Since CoCp(CO)₂ is easily removed, the thermal decomposition is expected to occur at low temperature and no self-saturation is expected. Note also that even with this low thermal stability, there is a limited self-saturation region between 1-3 s. This should be related to the existence of Cp ligand bonded to Co surface after CoCp(CO)₂ adsorption, but the weak saturation behavior is disturbed leading to CVD-like reaction at high dose of CoCp(CO)₂ leading to rapid increase in growth rate. However, the growth rate versus CoCp₂ exposure time (Fig. 3b) shows different behavior. The growth rate shows typical behavior of
ALD, with good saturation of growth rate at $t_s > 2$ s. The growth rate at saturation conditions was measured to be 0.48 Å/cycle. In contrast to CoCp(CO)$_2$, no deposition occurs without reactant. From these results, we can conclude CoCp$_2$ is a more proper precursor for Co ALD. This good saturation behavior of CoCp$_2$ is attributed to a fast reaction rate at saturation conditions, measured to be 0.48 Å/cycle. In the ALD saturation behavior is observed.

Figure 3 shows the resistivity of deposited Co thin films as a function of precursor exposure time. For PE-ALD of Co from CoCp$_2$ (Fig. 3b), the resistivity was low below 20 Ω cm for all range of exposure time, with the lowest resistivity of about 10 Ω cm, which is close to the bulk resistivity (6 Ω cm). This is one of the lowest values reported so far for vapor phase deposition of Co. For example, the resistivity of CVD Co from Co(CO)$_6$ and Co(CO)$_2$(NO) precursors was 11 and 15 Ω cm, respectively.

In contrast, the PE-ALD of Co from CoCp$_2$(Fig. 3a) shows low resistivity (20 Ω cm) at only limited exposure time range, where the ALD saturation behavior is observed.

Finally, Co silicide formation by annealing the PE-ALD Co film was studied. 20 nm thick PE-ALD Co films from CoCp$_2$ and NH$_3$ plasma with 20 nm thick PVD Ti capping layer were annealed by RTA for 30 s at various annealing temperature, $T_a$. The synchrotron radiation XRD spectra of PE-ALD Co on Si(001), annealed at $T_a$ between 400 and 900°C, are shown in Fig. 4 together with that of as-deposited PE-ALD Co. Basically, the silicide formation of PE-ALD Co with Ti cap layer follows similar trends as that of previous reported sputtered or evaporated Co films. 15, 16 While CoSi peaks are observed for annealing temperature between 500 and 600°C, clear CoSi$_2$(111) and (220) peaks are observed at above annealing temperature of 700°C.

In conclusion, Co PE-ALD processes were investigated using commercially available metallicorganic precursors and NH$_3$ plasma. The PE-ALD Co film has very low resistivity with high purity in contrast to thermal ALD Co using H$_2$ as a reactant. While PE-ALD process using CoCp(CO)$_2$ has shown incomplete self-saturation due to weak bonding of carbonyl ligand, CoCp$_2$ has shown good saturation behavior. The formation of CoSi$_2$ by post deposition annealing of PE-ALD Co with Ti cap layer was also successfully demonstrated.

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